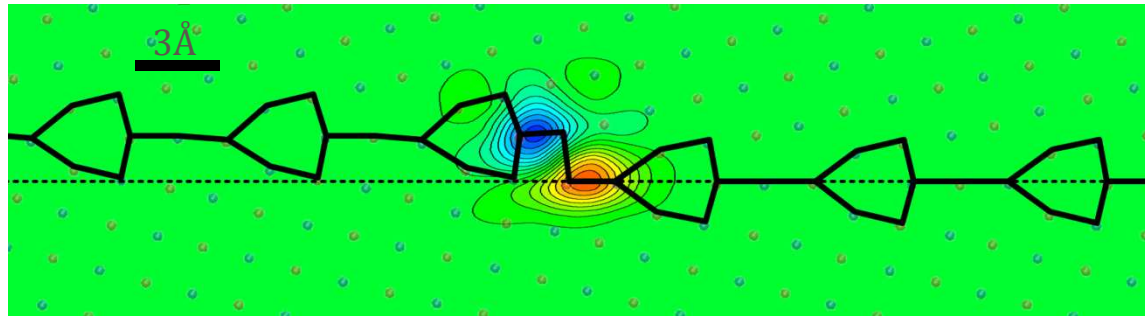


Challenges in continuous modeling of grain/phase boundaries



Generalized disclination dipole π_{113} in stepped tilt boundary, *X.Y. Sun et al. IJP 104, 134-146 (2018)*

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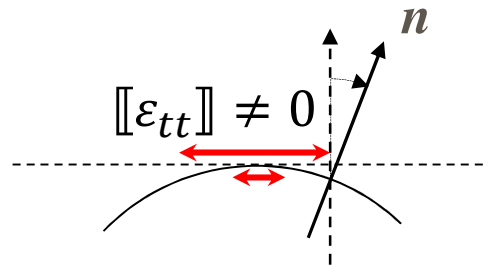
Motivations and modeling tools

■ Motivations

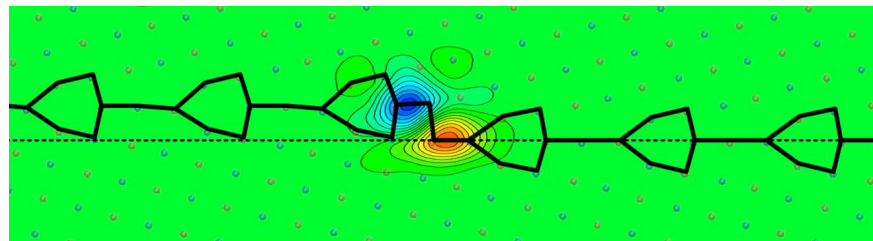
- ✓ Plasticity in nano-structured and slip-system-deprived materials (such as olivine)
- ✓ Structure/motion of « imperfect » interfaces (with curvature, steps and ledges)
- ✓ Boundary migration, grain growth, dynamic recrystallization, phase transformation

■ Tools for continuous interface modeling at various scales

- ✓ At microscale-mesoscale: Field Dislocation Mechanics +



- ✓ At nanoscale: Field Dislocation and Generalized Disclination Mechanics



■ Paradigm: converting discontinuities at interfaces into smooth fields

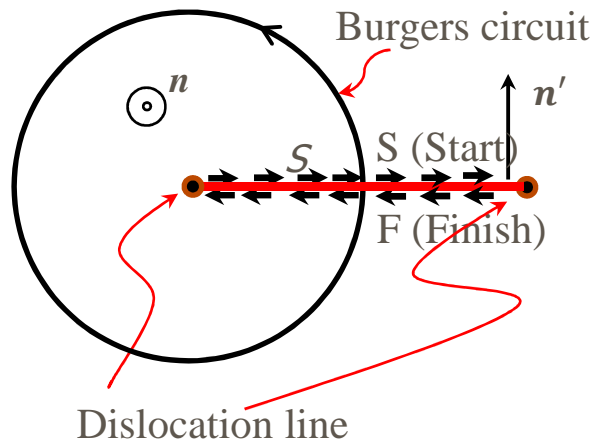
Outline

- Continuous description of interfaces at interatomic scale (FDGDM)
 - ✓ Crystal defect densities (dislocations/disclinations/generalized disclinations)
 - ✓ Elastostatic examples: Twin tips
 - ✓ Dynamic examples: Shear/Compression Coupled Boundary Migration
- Continuous description of interfaces at mesoscopic scale (FDM+)
 - ✓ Burgers vector conservation and tangential continuity of plastic distortion rate
 - ✓ Boundary curvature
- Conclusions
 - ✓ Relevance of continuity at nanoscale
 - ✓ Tangential continuity brings nonlocality to GB description at mesoscale
 - ✓ Spatial coarse graining

Continuous description of interfaces at interatomic scale

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Dislocation densities



V. Volterra, Ann. Sci. Ec. Norm. Sup. III 24, 401 (1907)

- **Discontinuity of elastic translational displacement** across a bounded surface

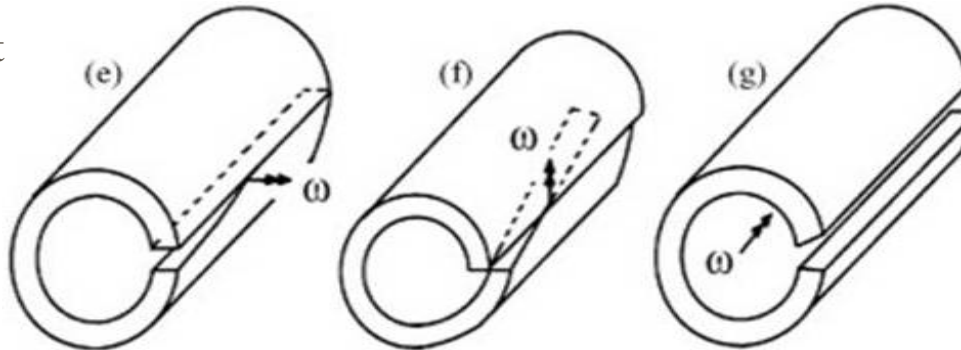
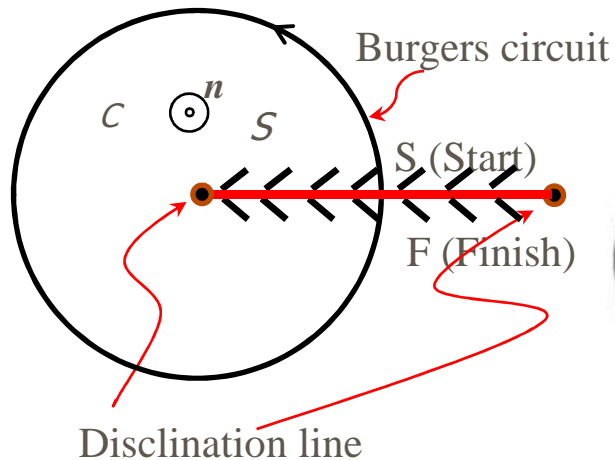
$$U_e = \cancel{\text{grad } u_e} \Rightarrow \llbracket u_e \rrbracket = b = \int_C U_e \cdot dx = \iint_S \text{curl } U_e \cdot n dS = \iint_S \alpha \cdot n dS \neq 0$$

- Incompatibility of the elastic distortion U_e and strain ϵ_e tensors reflected by Nye's dislocation density tensor α : E. Kröner, Erg. Ang. Math. 9, 1-179 (1958)

$$\text{curl } U_e = \text{curl } \epsilon_e + \text{curl } \omega_e = \text{curl } \epsilon_e + \text{tr}(\kappa_e)I - \kappa_e^t = \alpha$$

- **Continuity of elastic rotation tensor, tangential continuity of elastic distortion and strain tensors:** $\kappa_e = \text{grad } \omega_e$, $\llbracket U_e \rrbracket \times n' = 0$, $\llbracket \epsilon_e \rrbracket \times n' = 0$

Disclination densities



V. Volterra, Ann. Sci. Ec. Norm. Sup. III 24, 401 (1907)

- **Discontinuity of elastic rotation** across a bounded surface

$$\cancel{\kappa_e = \text{grad } \omega_e} \Rightarrow [[\omega_e]] = \Omega = \int_C \kappa_e \cdot dx = \iint_S \text{curl } \kappa_e \cdot ndS = \iint_S \theta \cdot ndS$$

- Incompatibility of the elastic curvature tensor κ_e reflected by deWit's disclination density tensor θ :

$$\text{curl } \kappa_e = \theta$$

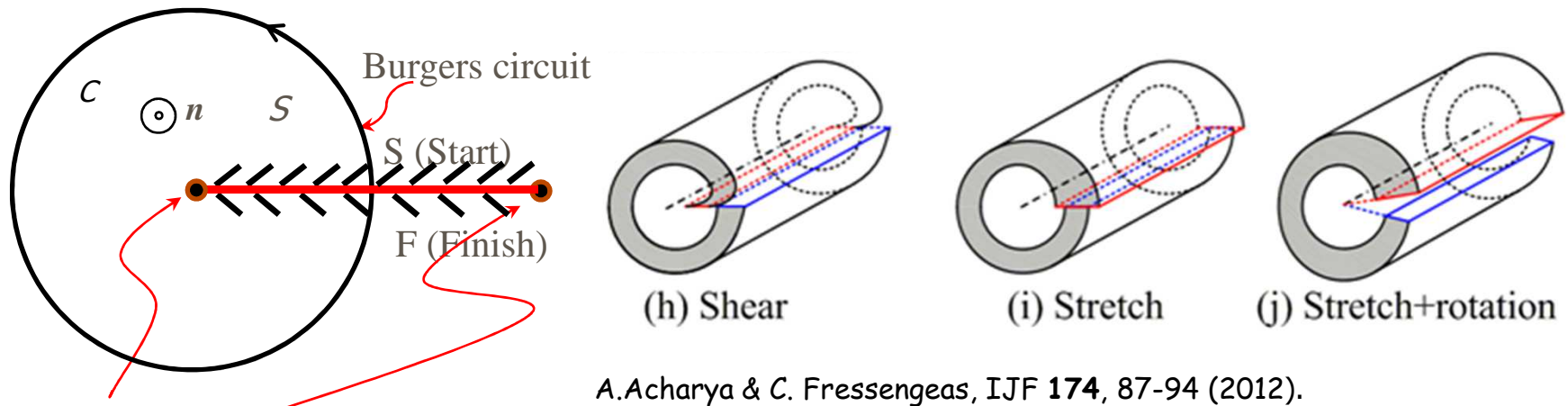
R. deWit, JRNBS-A 5, 651-680 (1970)

- **Tangential continuity of the elastic curvature, distortion and strain tensors**

$$[[U_e]] \times n' = 0, [[\varepsilon_e]] \times n' = 0, [[\kappa_e]] \times n' = 0$$

C. Fressengeas & B. Beausir, IJSS 156-157, 210-215 (2019)

Generalized disclination densities



- **Discontinuity of elastic distortion** across a bounded surface

$$\cancel{G_e = \text{grad } U_e} \Rightarrow [[U_e]] = \Pi = \int_C G_e \cdot dx = \iint_S \text{curl } G_e \cdot ndS = \iint_S \pi \cdot ndS$$

- Incompatibility (non-integrability) of the second-order distortion G_e reflected by the g-disclination density tensor π :

$$\boxed{\text{curl } G_e = \pi}$$

Generalized disclination dipoles in straight tilt boundaries

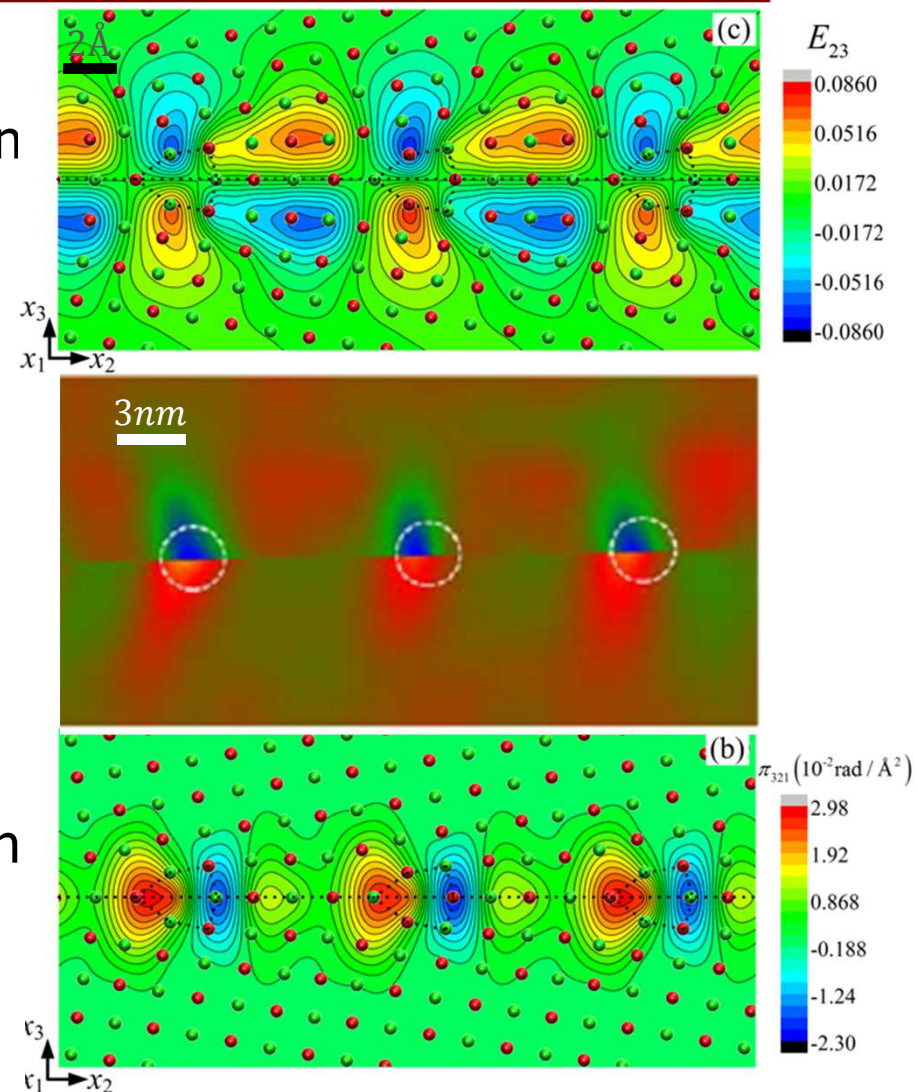
- Shear discontinuity $[[\varepsilon_{23}]]$ accommodated in less than 2 Å in copper $\Sigma 37(610)$ symmetric $\langle 100 \rangle$ tilt boundary (18.9°)

X.Y. Sun et al, IJP 77 75-89 (2016)

- Shear discontinuity $[[\varepsilon_{23}]]$ in silicon $\Sigma 9 \langle 122 \rangle$ 38.94° tilt boundary from Geometric Phase Analysis

M. Couillard et al, Phil.Mag. (2013)

- G-disclination density field π_{231} reflecting discontinuity of rotation $[[\omega_{23}]]$ and shear strain $[[\varepsilon_{23}]]$ in copper $\Sigma 37(610)$ symmetric $\langle 100 \rangle$ tilt boundary (18.9°)



Generalized disclination dipoles in stepped tilt boundaries

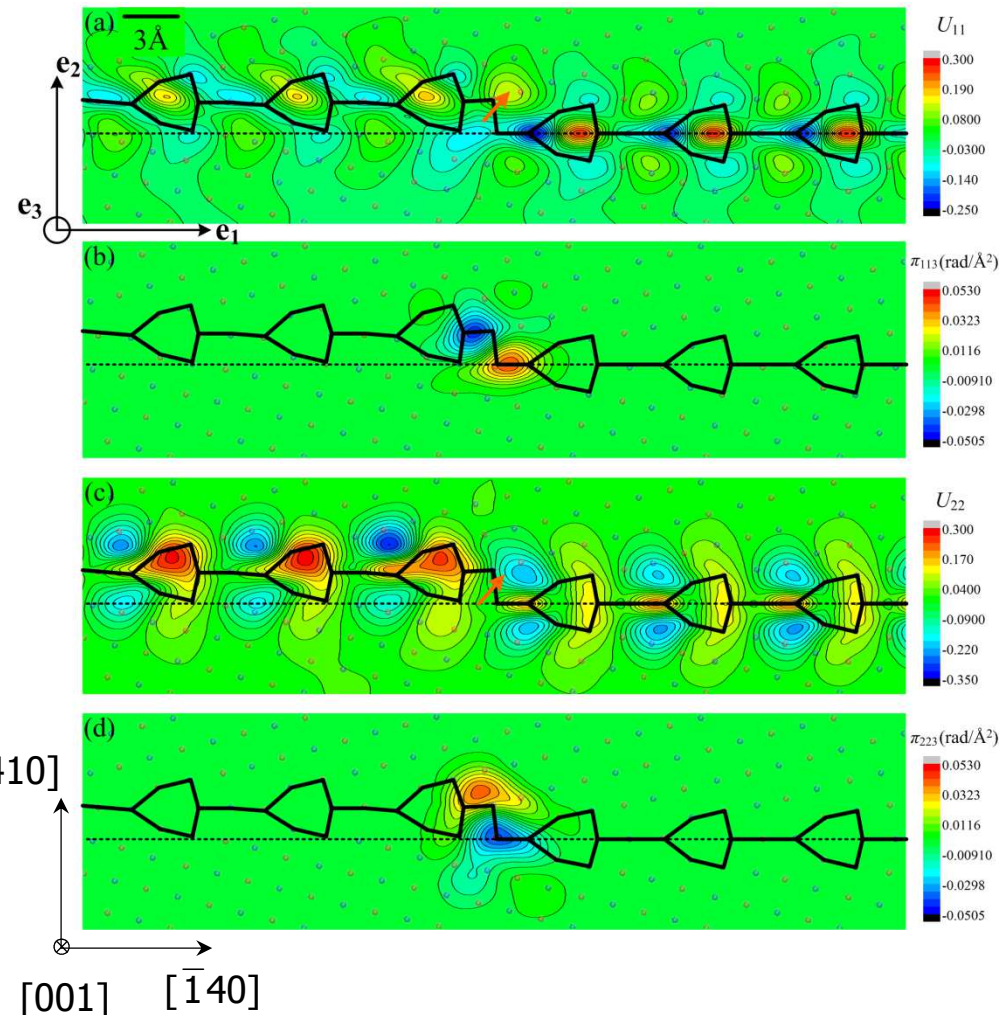
- $[[\epsilon_{11}]]$ discontinuity in $\langle 100 \rangle$ ledge of tilt boundary $\Sigma 17(410)$, 28.07°

- π_{113} g-disclination dipole

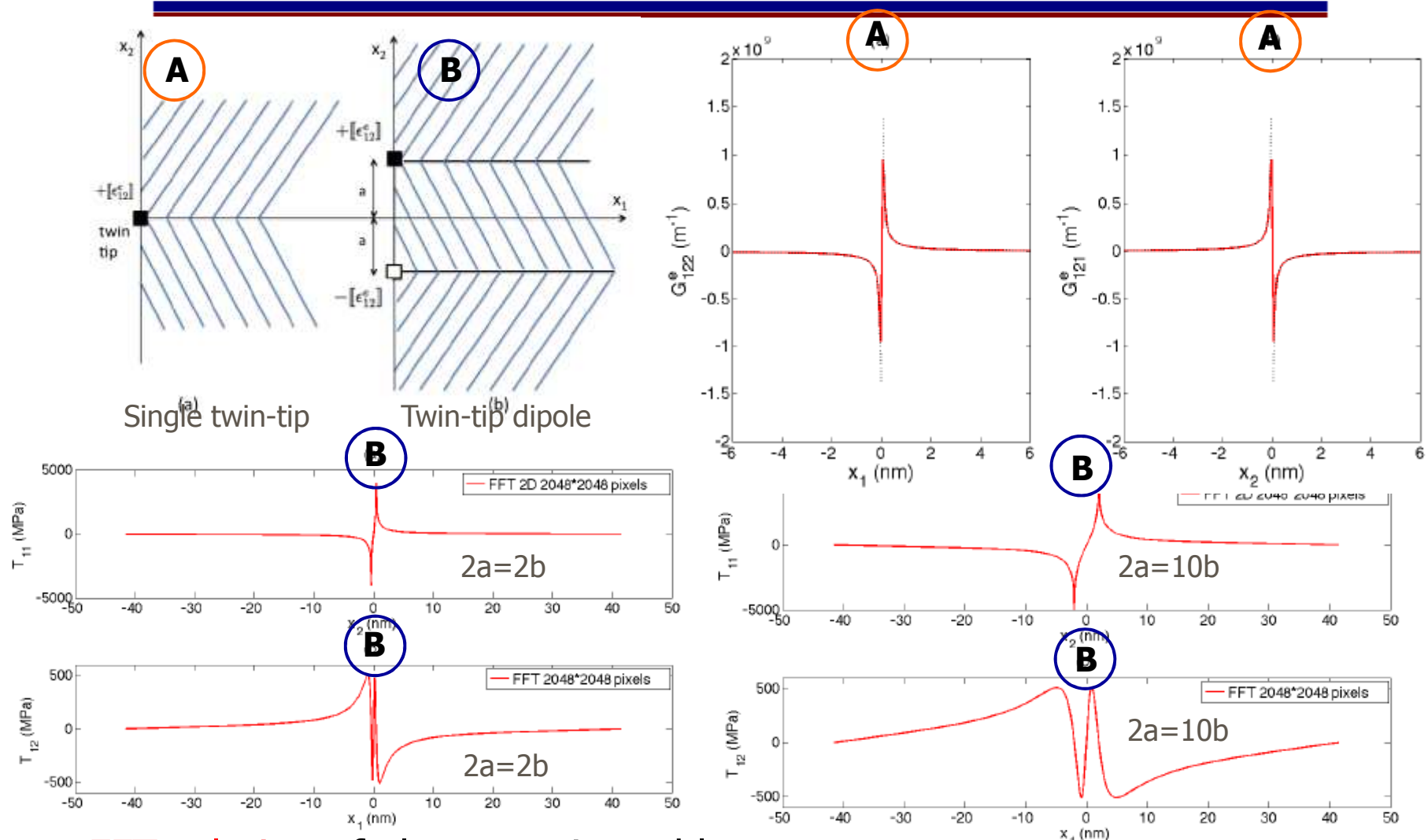
- $[[\epsilon_{22}]]$ discontinuity in $\langle 100 \rangle$ ledge of tilt boundary $\Sigma 17(410)$, 28.07°

- π_{223} g-disclination dipole

X.Y. Sun et al., IJP **104**, 134-146 (2018)

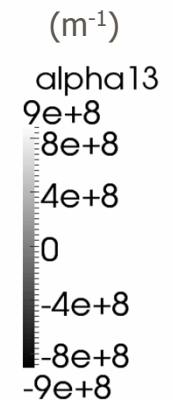
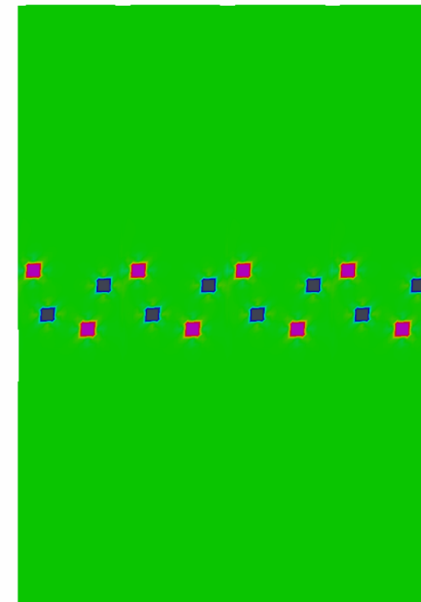
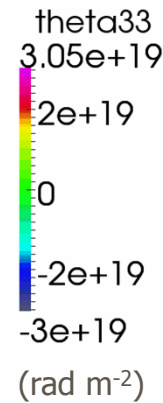
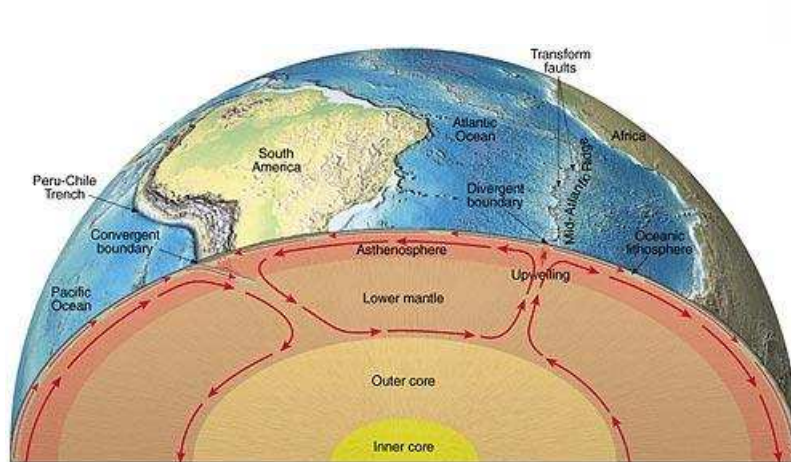


Elasto-static fields of twin-tips



■ FFT solution of elasto-static problem: Berbenni et al, IJSS 51, 4157-4175 (2014)

Shear Coupled Boundary Migration in olivine

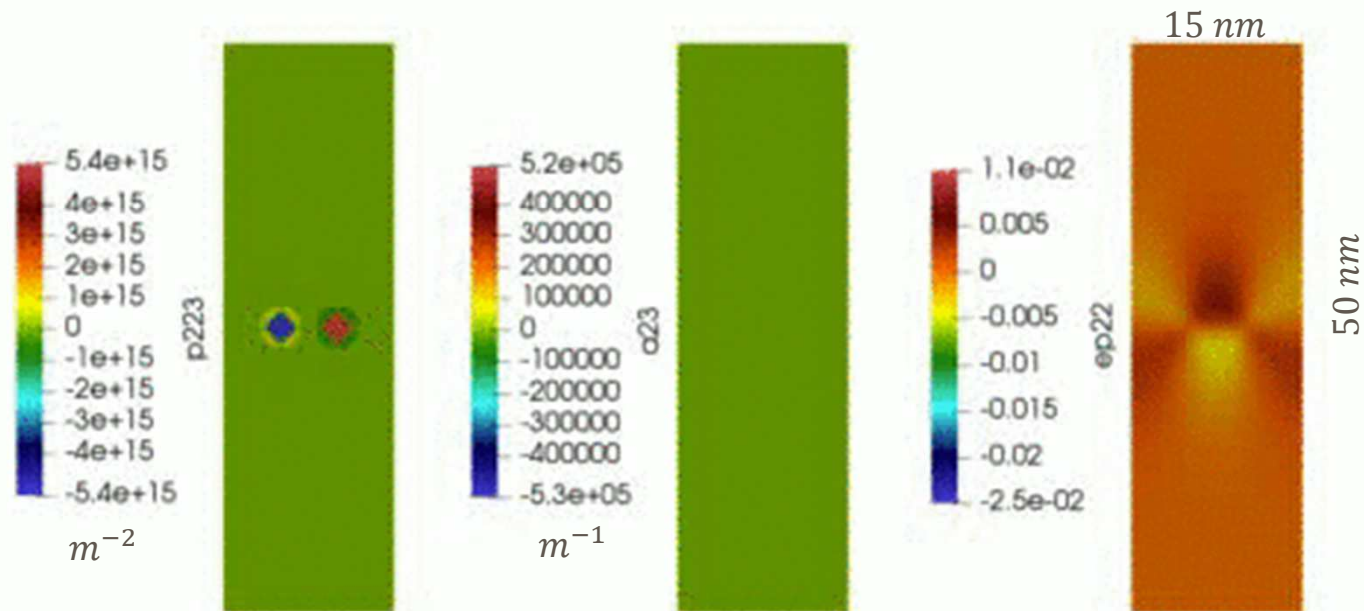


- Olivine (Mg,Fe)₂SiO₄: main constituent of the Earth's upper mantle (~70%)
- Orthorhombic olivine has only three independent slip systems
- Dislocation glide cannot accommodate mantle's convective motion
- FEM solution of FDDM field equations
- Shear-coupled boundary migration provides an additional deformation mechanism for slip system-deprived olivine

P. Cordier et al., *Nature*, **507**, 51-56 (2014).

Compression induced boundary migration

- FEM solution of FDGDM field equations
- Migration of π_{223} dipoles generates α_{23} dipoles
- Tangential climb of α_{23} induces normal compression $\dot{U}_{22}^p = \alpha_{23}V_1$
- Normal glide of α_{23} induces shear + rotation $\dot{U}_{21}^p = -\alpha_{23}V_2$



TimeMan, 6 Février 2019

V. Taupin & C. Fressengeas, CSMA 2019

Continuous description of interfaces at mesoscopic scale

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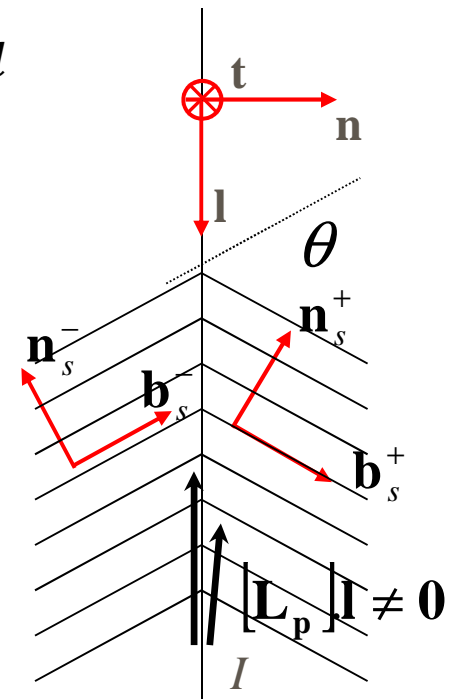
Interfaces in conventional CP: tangential plastic distortion rate discontinuity

- Typical example: symmetric tilt boundary
 - ☞ Discontinuity of the elastic rotation: $[[\omega_e]] = \theta \mathbf{n} \times \mathbf{l}$
- Tangential discontinuity of plastic distortion rate

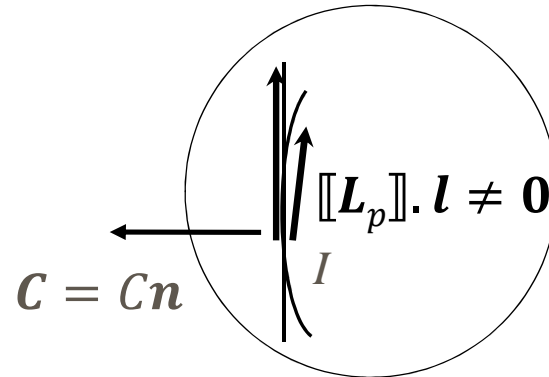
$$\forall \mathbf{l} \in I, [[\mathbf{L}_p]] \cdot \mathbf{l} = \left(\sum \dot{\gamma}_s \mathbf{b}_s^+ \otimes \mathbf{n}_s^+ - \sum \dot{\gamma}_s \mathbf{b}_s^- \otimes \mathbf{n}_s^- \right) \cdot \mathbf{l} \neq 0$$

- Any discontinuity $[[\mathbf{L}_p]]$ is accepted and (sometimes) accommodated through Frank-Bilby *surface-dislocation* densities in interface

$$\forall \mathbf{l} \in I, [[\mathbf{L}_p]] \cdot \mathbf{l} = \dot{\alpha}^S(I) \cdot \mathbf{t} \neq 0$$

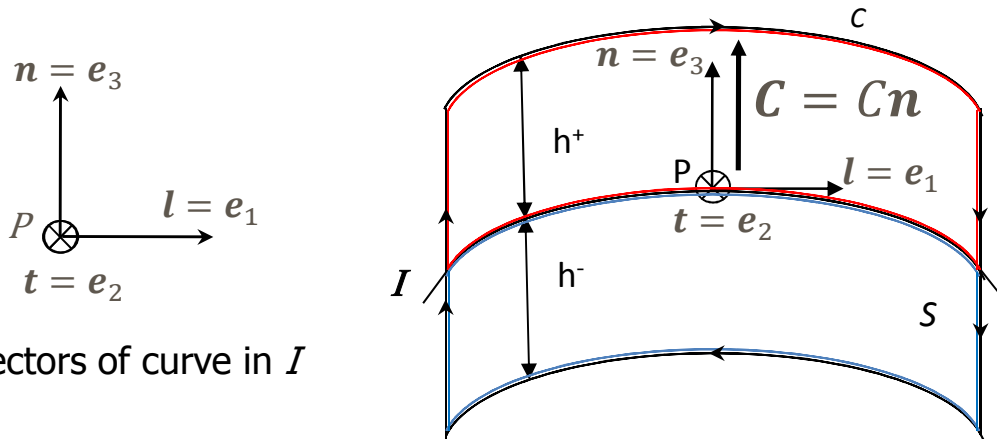


Burgers vector conservation and curvature dynamics



- Ban surface-dislocations $\alpha^S(I)$
- Distribute surface-dislocations in a volumetric layer through tangential continuity of plastic distortion
V. Taupin et al, IJSS **99**, 71-81 (2016)
- Relate tangential plastic distortion rate discontinuity with interface curvature dynamics through GB-mediated plasticity
C. Fressengeas, JMPS **124**, 814-826 (2019)

Continuity constraints on plastic distortion rate on curving interfaces



Darboux directors of curve in I

- Conservation of Burgers vector:

$$\forall l \in I, \dot{\mathbf{b}} = \frac{d}{dt} \iint_S \boldsymbol{\alpha} \cdot \mathbf{t} dS = - \frac{d}{dt} \int_C \mathbf{U}_p \cdot \mathbf{l} ds = 0$$

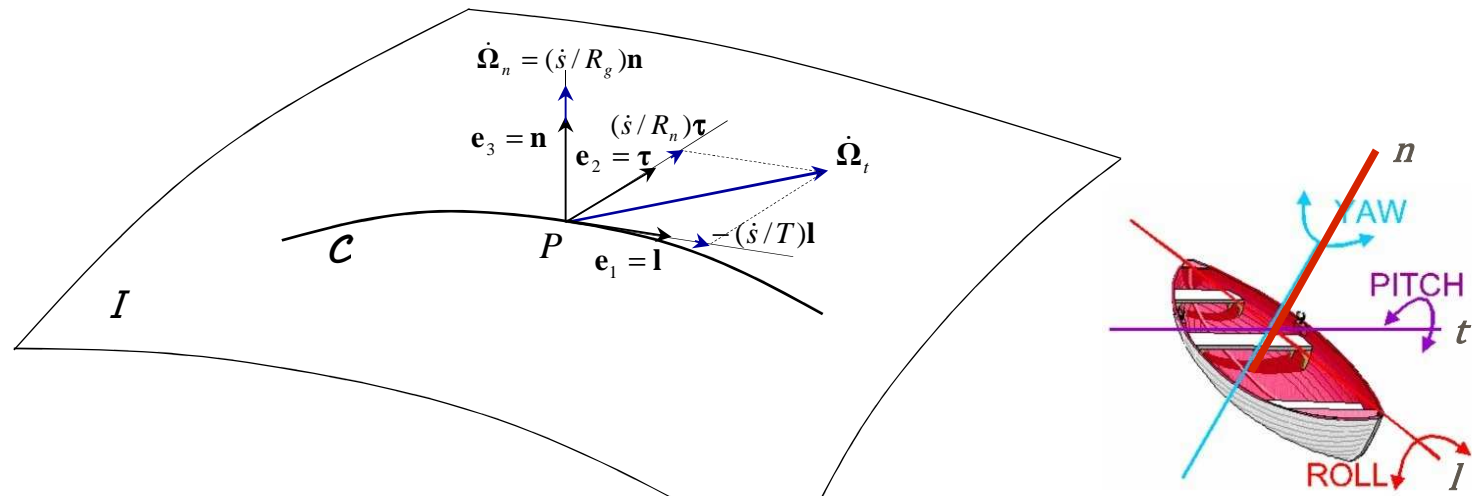
- Allow **rotation of interface directors**; in the limit $h^+, h^- \rightarrow 0, L \rightarrow 0$,

$$\forall l \in I, (\llbracket \dot{\mathbf{U}}_p \rrbracket - \llbracket \boldsymbol{\alpha} \rrbracket \times \mathbf{C}) \cdot \mathbf{l} + \llbracket \mathbf{U}_p \rrbracket \cdot \dot{\mathbf{l}} = 0$$

$$\dot{\mathbf{l}} = \dot{\boldsymbol{\Omega}} \times \mathbf{l}, \quad \dot{\boldsymbol{\Omega}} = \left(-\frac{\mathbf{l}}{T} + \frac{\mathbf{t}}{R_n} - \frac{\mathbf{n}}{R_g} \right) \dot{s}$$

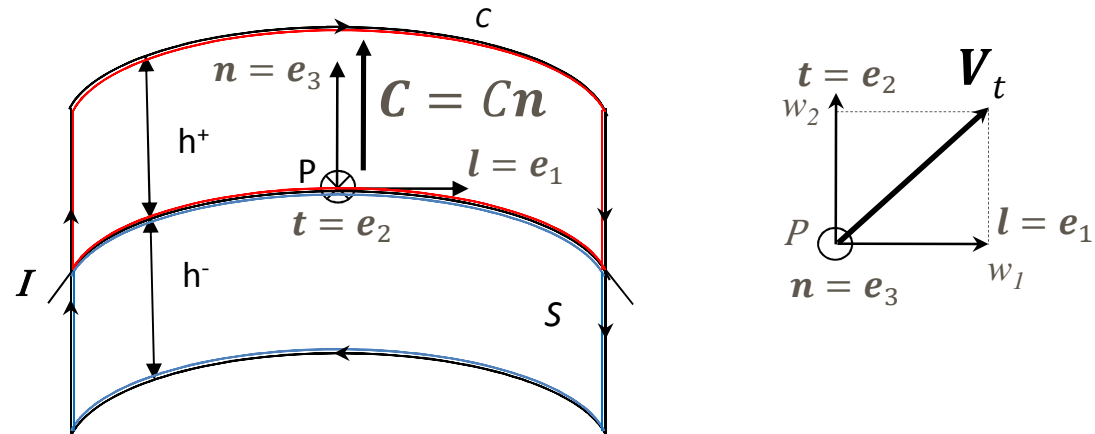
- ☞ Darboux' frame rotation components: $1/R_g, 1/R_n, 1/T$, geodesic curvature, normal curvature and geodesic torsion

Darboux frame



- $\forall i, \dot{e}_i = \dot{\Omega} \times e_i$, $\dot{\Omega} = \left(-\frac{l}{T} + \frac{t}{R_n} - \frac{n}{R_g} \right) \dot{s}$ in rad.s^{-1} : rotation rate of Darboux frame. $1/T$ is the geodesic torsion (roll), $1/R_n$ the normal curvature (pitch), $1/R_g$ the geodesic curvature (yaw).
- Along principal directions $1/T = 0$, normal curvature = principal curvature; geodesic curvature $1/R_g$ not used in the model

Continuity constraints on plastic distortion rate on curving interfaces



- Plastic distortion rate discontinuity in the limit $h^+, h^- \rightarrow 0, L \rightarrow 0$,

$$\forall l \in I, (\llbracket \dot{U}_p \rrbracket - \llbracket \alpha \rrbracket \times C) \cdot l + \llbracket U_p \rrbracket \cdot \dot{\Omega} \times l = 0$$

$$\Leftrightarrow (\llbracket \alpha \rrbracket \times (V - C) + \llbracket U_p \rrbracket \times \dot{\Omega}) \times n = 0$$

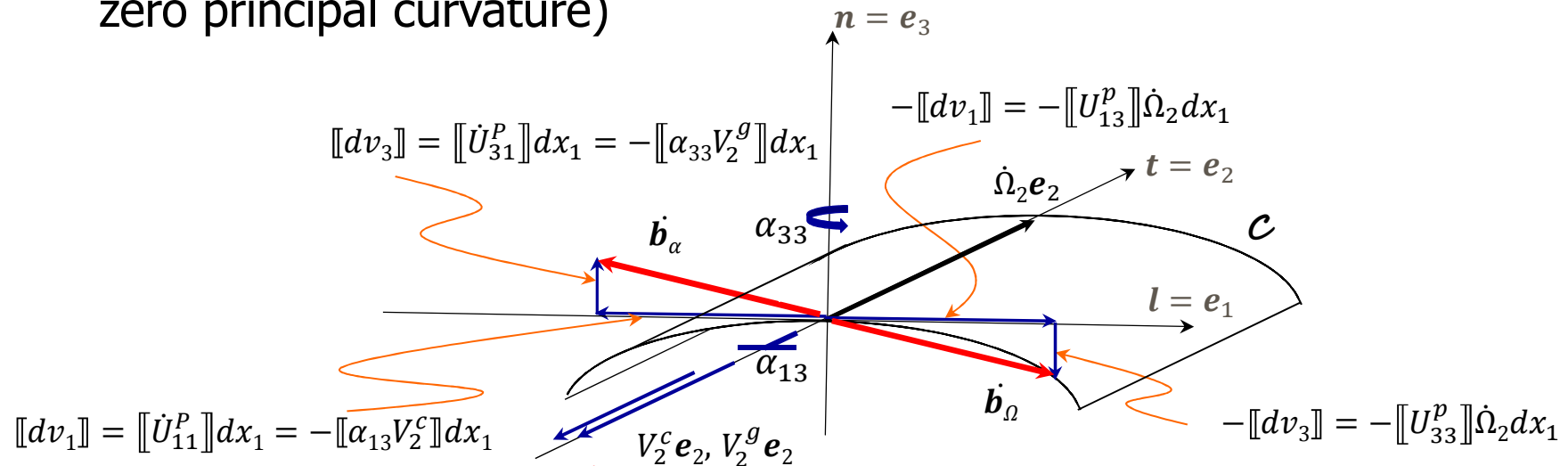
$$\Leftrightarrow \llbracket \alpha \cdot n \otimes V_t - (V_n - C_n) \alpha_t \rrbracket + \llbracket U_p \rrbracket \cdot n \otimes \dot{\Omega}_t = 0$$

- Interface rotation rate offsets tangential discontinuity of plastic distortion rate

$$\dot{\Omega}_t = -\llbracket \alpha \cdot n \otimes V_t - (V_n - C_n) \alpha_t \rrbracket^t \cdot \frac{\llbracket U_p \rrbracket \cdot n}{(\llbracket U_p \rrbracket \cdot n)^2}$$

Cylindrical example

- Principal directions in cylindrical interface (no torsion, only one non-zero principal curvature)



- Interface rotation rate $\dot{\Omega}_2$ offsets discontinuities arising from **axial climb** V_2^c of α_{13} edges and **glide** V_2^g of α_{33} screws

$$\begin{aligned} [[\dot{U}_{11}^p]] &= -[[\alpha_{13} V_2^c]] = \dot{\Omega}_2 [[U_{13}^p]] \Leftrightarrow \dot{b}_1^\alpha = -\dot{b}_1^\Omega \\ [[\dot{U}_{31}^p]] &= -[[\alpha_{33} V_2^g]] = \dot{\Omega}_2 [[U_{33}^p]] \Leftrightarrow \dot{b}_3^\alpha = -\dot{b}_3^\Omega \end{aligned}$$

- Curvature dynamics: $\dot{\Omega}_2 = -[[\alpha_{33} V_2^g]] / [[U_{33}^p]]$

C. Fressengeas, JMPS 124, 814-826 (2019)

Conclusions

- Continuous modeling of GB elastic structure at nanoscale
 - ✓ Periodic arrays of g-disclination dipoles compose GB core structure
 - ✓ Boundary migration for realistic stress and strain rate fields, numerical efficiency
- At mesoscopic scale, nonlocality is brought to interface modeling by Burgers vector conservation and tangential continuity conditions
- Coarse-graining eased by unique overall structure of the theory
- Related features (not shown here)
 - ✓ Nonlocality, nonlinearity of elasticity in defects cores: V. Taupin et al, *JMPS* 100, 62-84 (2017)
 - ✓ Prediction of interface velocity in mesoscopic model: C. Fressengeas, *JMPS* 124, 814-826 (2019)
- Potential use for prediction of
 - ✓ Migration of imperfect GBs in both FDM+ and FDGDM
 - ✓ Cracks-extrinsic defects-GB interactions in FDGDM